# Examination of Capacities of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> Nanocages as High Efficiency Catalysts for Oxygen Reduction Reaction (ORR) in Fuel Cell

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Summary: Here, the capacities of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages of ORR are examined. The mechanisms of ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages as catalysts are investigated. Results indicated that Cr have favorable bonds with Si, C and Ge of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages. The OOH adsorption on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages is more favorite process than O<sub>2</sub> dissociation from thermodynamic view point. The \*OH formation is rate-determining for ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages. The Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> have higher catalytic activity for ORR processes than metal-based catalysts and AlN, C, BN, AlP and BP nanocages and AlN, C, BN, Si, Ge, AIP and BP nanotubes. The over-potential of ORR on surfaces of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge60 are lower than metal based catalysts. The Cr-Si60 and Cr-Ge60 nanocages have lower overpotential for ORR than Cr-C<sub>60</sub> nanocage. The Cr-Si<sub>60</sub> and Cr-Ge<sub>60</sub> are proposed as new catalysts for reaction steps of ORR by acceptable performance.

**Keyword:** Oxygen-RR, Si<sub>60</sub>, Nano-catalyst, Reduction mechanism, Acid environment, Over-potential.

### Introduction

The oxygen reduction reaction (ORR) can be reduced the efficiency and performance of fuel cells [1]. The one way to increase the efficiency of fuel cells is to reduce the oxygen reduction reaction (ORR). Researchers have made great efforts to find suitable catalysts for the oxygen ORR. The metal-based catalysts have moderate efficiency for the ORR [2]. The metalbased catalysts have low stability and high cost, as well as environmental problems for the ORR [3].

The carbon nanoparticle-based catalysts for the ORR have high stability, low cost, and large surface area [4]. The carbon nanoparticle-based catalysts can improve the efficiency of fuel cells by catalyzing the ORR [5]. The adsorption of metals on the surface of carbon-based catalysts can significantly increase the efficiency of carbon nanoparticle-based catalysts. The metals adsorbed on carbon nanoparticle-based catalysts have the ability to adsorb species related to ORR [6].

In previous works, the capacities of various metal-based catalysts [7] and Metals (Fe, Co, Ni and Cu) doped AlN, C, BN, AlP and BP nanocages and Metals (Fe, Co, Ni and Cu) doped AlN, C, BN, Si, Ge, AlP and BP nanotubes [8] for ORR processes have been examined. The mechanisms for ORR processes on surfaces of various metal-based catalysts [9] and Metal doped nanostructures have been investigated [10]. Results have shown than the Metal doped nanostructures [11] have higher catalytic activity for ORR processes than various metal-based catalysts [12]. Results have indicated that the pathways and their steps of ORR on various metal-based catalysts [13] and Metal doped nanostructures have been investigated [14] are same.

Researcher have demonstrated that the C<sub>60</sub> structure is one of the most stable nanocages that has high surface area, thermodynamic stability and high catalytic activity for inorganic reactions [15]. The other derivatives of C<sub>60</sub> structure such as Si<sub>60</sub> and Ge<sub>60</sub> structures similar to C<sub>60</sub> structure have high thermodynamic stability and formation energy and their surfaces area have higher than  $C_{60}$  structure. The metal adoption on surfaces of C<sub>60</sub>, Si<sub>60</sub> and Ge<sub>60</sub> structures can improve their catalytic capacities to adsorb the intermediates of important organic reactions [16].

In this study, the potential of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages of ORR are investigated. The effects of adsorption of Cr on  $Si_{60}$ ,  $C_{60}$  and  $Ge_{60}$  on capacities for ORR are examined. The acceptable pathways of ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> are examined. The goals of this work are: to suggest the catalysts (Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub>) for ORR with acceptable performances; to compare the performances of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> for ORR; to propose available pathways of ORR on Cr-

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Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub>; to find the Cr adsorption effects on capacities of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> for ORR; to compare the efficiency of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> for ORR with metal based catalysts.

### Computational details

The geometries of Si<sub>60</sub>, C<sub>60</sub>, Ge<sub>60</sub>, Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> and complexes with ORR derivatives have been optimized by M06-2X/6-311+G (2d, 2p) model in GAMESS software [16, 17]. The frequencies of Si<sub>60</sub>, C<sub>60</sub>, Ge<sub>60</sub>, Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> and complexes with ORR derivatives are calculated to confirm that the nano-structures are stable structures [18, 19]. The capacities of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> to adsorb of ORR species are examined. The structures of  $Si_{60}$ ,  $C_{60}$  and  $Ge_{60}$  nanocages are reported in Figure 1. The E<sub>formation</sub> of Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> are summarized in Figure 1. The possible mechanisms for ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages are examined to propose the acceptable mechanism and new catalysts with high performances. The ORR on surfaces of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages is processed by three pathways as presented in Table 1. The  $\Delta G_{reaction}$  and  $E_{activation}$  of ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> are investigated in Table

## Results and discussion

Adsorption of ORR species on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge60 nanocages

The  $E_{\text{formation}}$  of  $Si_{60}$ ,  $C_{60}$  and  $Ge_{60}$  nanocages are -5.89, -4.42 and -5.58 eV. The E<sub>formation</sub> of Si<sub>60</sub> are higher than C<sub>60</sub> and Ge<sub>60</sub> nanocages ca 1.52 and 0.31 eV. Results shown that E<sub>formation</sub> of Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> nanocages are negative.

The  $E_{formation}$  of  $Si_{60}$ ,  $C_{60}$  and  $Ge_{60}$  are calculated as following [19]:

$$E_{\text{formation}} = E_{\text{nanocage}} - 60 * E_{\text{X}}$$
 (1)

The E<sub>nanocage</sub> is total energies of Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> nanocages and the E<sub>X</sub> are energies of Si, C, Ge atoms. The adoption energies (E<sub>adoption</sub>) of Cr atoms on Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> nanocages are calculated as following [20]:

$$E_{adoption} = E_{Cr-nanocage} - E_{nanocage} - E_{Cr}$$
 (2)

The E<sub>Cr-nanocage</sub> is total energies of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> complexes and the E<sub>nanocage</sub> is total energies of Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> and E<sub>Cr</sub> is energy of Cr. The E<sub>adsorption</sub> of ORR derivatives on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages are calculated as following [21]:

$$E_{adsorption} = E_{specie-Cr-nanocage} - E_{Cr-nanocage} - E_{specie}$$
 (3)

The E<sub>specie-Cr-nanocage</sub> is energies of complexes of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages with ORR derivatives, the E<sub>Cr-nanocage</sub> is energies of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages and the E<sub>specie</sub> is energies of ORR derivatives.

The Cr adoption on Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> nanocages are examined and structures of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> are presented in Figure 1. The E<sub>adoption</sub> of Cr on Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> are shown in Figure 1. The E<sub>adoption</sub> of Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> nanocages are -4.56, -4.15 and -4.41 eV. The E<sub>adoption</sub> of Cr-Si<sub>60</sub> are more negative than Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages ca 0.41 and 0.15 eV. The E<sub>adoption</sub> of Cr on Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> are negative and Cr have acceptable bonds with Si, C and Ge of Cr-Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> nanocages. The Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> are stable structures and Cr is active site for ORR adsorption.

In this section, the adsorption of ORR species on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> are investigated. The structures of complexes of ORR species with Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> are shown in Figure 1. The O<sub>2</sub> is joined to Cr of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages. The E<sub>adsorption</sub> of O<sub>2</sub> on Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> nanocages are -0.45, -0.37 and -0.40 eV. The E<sub>adsorption</sub> of O on Si<sub>60</sub>, C<sub>60</sub> and  $Ge_{60}$  nanocages are -3.30, -2.87 and -2.96 eV. The E<sub>adsorption</sub> of O + O on Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> nanocages are -6.54, -5.78 and -5.87 eV.

The O+O and O have high interactions with Cr of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> and have acceptable  $E_{adsorption}$ . The  $O_2$  on Cr- $Si_{60}$ , Cr- $C_{60}$  and Cr- $Ge_{60}$ nanocages is produced the O and O + O on Cr of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages. The E<sub>adsorption</sub> values of O +O and O with Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> are higher than O2. The OH and OOH are adsorbed on Cr atoms of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages and E<sub>adsorption</sub> are higher than O<sub>2</sub>. The Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> have higher capacities for ORR species adsorption than OH and OOH. The H<sub>2</sub>O on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> are physically absorbed and H<sub>2</sub>O is easily desorbed from Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages. The E<sub>adsorption</sub> of ORR species on Cr-Si<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages are higher than Cr-C<sub>60</sub> nanocage.

In this study, the  $E_{HLG}$  and q as charge transfer of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages and complexes of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> with ORR species are reported in Table 2. The complexes of ORR species (O. OH, OOH, O + O and  $H_2O$ ) on Cr-Si<sub>60</sub> have lower  $E_{HLG}$ than corresponding values for Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages. The q as charge transfer between the ORR species (O, OH, OOH, O + O and H<sub>2</sub>O) and Cr-Si<sub>60</sub> have higher corresponding values for Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages. The Cr-Si<sub>60</sub> and Cr-Ge<sub>60</sub> have higher capacities to join with ORR species than Cr-C<sub>60</sub>. The Cr-Si<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages have higher potential than Cr-C<sub>60</sub> for ORR.

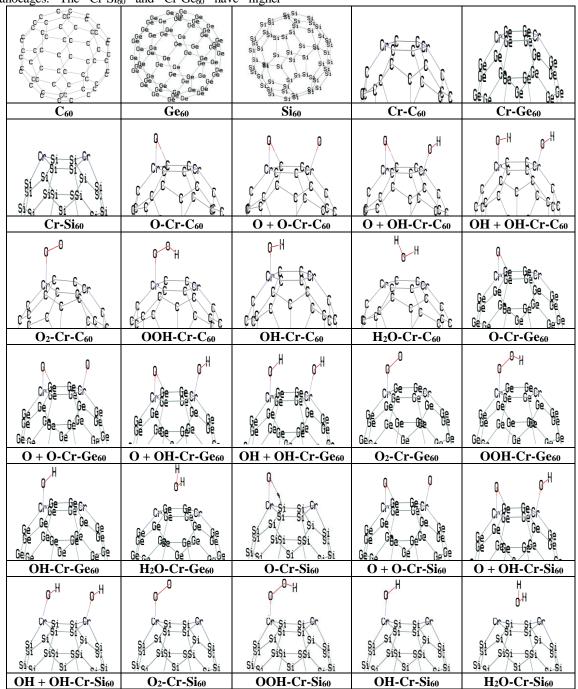


Fig. 1:Structures of Si<sub>60</sub>, C<sub>60</sub> and Ge<sub>60</sub> nanocages, Cr doped nanocages (Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub>) and complexes with ORR species.

Table-1: The Faction and AGreeties in eV for ORR on Cr-Sign Cr-Co and Cr-Gego nanocages

Table-1: The $E_{activation}$ and $\Delta G_{reaction}$ in eV for OR	R on Cr-S1 <sub>60</sub>	, Cr-C <sub>60</sub> ar	nd Cr-Ge <sub>6</sub>	<sub>0</sub> nanocage	S.		
	11+G (2d, 2p) mo						
Nanocages	C	$Cr-C_{60}$ $Cr-Ge_{60}$		Cr-Si <sub>60</sub>			
ORR pathways 1, 2 and 3	$\mathbf{E}_{\mathbf{act}}$	$\Delta G_{reac}$	$\mathbf{E}_{\mathbf{act}}$	$\Delta G_{reac}$	$\mathbf{E}_{\mathbf{act}}$	$\Delta G_{reac}$	
$surface\text{-}{}^*O_2 \to {}^*O\text{-}surface\text{-}O^*$	1.30	-1.52	1.34	-1.56	1.19	-1.74	
surface-*O2 → surface-*OOH	0.44	-1.39	0.45	-1.42	0.40	-1.58	
surface-*OOH → surface-O*	1.12	-2.68	1.14	-2.75	1.02	-3.06	
surface-O* → surface-*OH	1.05	-1.70	1.08	-1.74	0.96	-1.94	
$surface *OH \rightarrow surface + H_2O$	0.66	-0.60	0.67	-0.62	0.60	-0.69	
$surface-*OOH \rightarrow HO*-surface-*OH$	1.55	-3.38	1.59	-3.46	1.42	-3.85	
HO*-surface-*OH → surface-*OH + H <sub>2</sub> O	0.96	-1.81	0.98	-1.86	0.87	-2.07	
$surface *OOH \rightarrow HO*-surface *O*$	1.01	-1.17	1.04	-1.20	0.93	-1.34	
HO*-surface-O* → surface-O* + H <sub>2</sub> O	1.13	-1.45	1.15	-1.49	1.02	-1.66	
surface-O* → surface-*OH	1.05	-1.69	1.08	-1.73	0.96	-1.93	
	1+G (2d, 2p) mo						
Nanocages		r-C <sub>60</sub>		-Ge <sub>60</sub>	Cr-Si <sub>60</sub>		
ORR pathways 1, 2 and 3	$\mathbf{E}_{\mathbf{act}}$	$\Delta G_{reac}$	$\mathbf{E}_{\mathrm{act}}$	$\Delta G_{reac}$	$\mathbf{E}_{\mathbf{act}}$	$\Delta G_{reac}$	
$surface \text{-}{}^*O_2 \rightarrow \text{*}O\text{-}surface \text{-}O^*$	1.33	-1.49	1.37	-1.52	1.22	-1.70	
$surface-*O_2 \rightarrow surface-*OOH$	0.45	-1.36	0.46	-1.39	0.41	-1.54	
surface-*OOH → surface-O*	1.15	-2.62	1.17	-2.69	1.04	-2.99	
$surface-O^* \rightarrow surface-*OH$	1.07	-1.66	1.10	-1.70	0.98	-1.90	
surface $*OH \rightarrow surface + H_2O$	0.68	-0.59	0.69	-0.61	0.61	-0.67	
$surface-*OOH \rightarrow HO*-surface-*OH$	1.59	-3.30	1.63	-3.38	1.45	-3.76	
HO*-surface-*OH → surface-*OH + H <sub>2</sub> O	0.98	-1.77	1.00	-1.82	0.89	-2.02	
surface-*OOH $\rightarrow$ HO*-surface-O*	1.03	-1.14	1.06	-1.17	0.95	-1.31	
HO*-surface-O* → surface-O* + H <sub>2</sub> O	1.16	-1.42	1.18	-1.46	1.04	-1.62	
surface-O* → surface-*OH	1.07	-1.65	1.10	-1.69	0.98	-1.89	
C	OSMO model in	water					
Nanocages	C	Cr-C <sub>60</sub>		Cr-Ge <sub>60</sub>		Cr-Si <sub>60</sub>	
ORR pathways 1, 2 and 3	$\mathbf{E}_{\mathbf{act}}$	$\Delta G_{reac}$	$\mathbf{E}_{\mathbf{act}}$	$\Delta G_{reac}$	$\mathbf{E}_{\mathbf{act}}$	$\Delta G_{reac}$	
$surface \text{-} *O_2 \rightarrow *O\text{-}surface \text{-}O*$	1.27	-1.55	1.31	-1.60	1.16	-1.78	
$surface-*O_2 \rightarrow surface-*OOH$	0.43	-1.42	0.44	-1.45	0.39	-1.62	
surface-*OOH → surface-O*	1.09	-2.74	1.11	-2.81	1.00	-3.13	
$surface-O^* \rightarrow surface-*OH$	1.03	-1.74	1.06	-1.78	0.94	-1.98	
surface-*OH $\rightarrow$ surface + H <sub>2</sub> O	0.65	-0.61	0.65	-0.63	0.59	-0.71	
$surface-*OOH \rightarrow HO*-surface-*OH$	1.52	-3.46	1.55	-3.54	1.39	-3.94	
HO*-surface-*OH → surface-*OH + H <sub>2</sub> O	0.94	-1.85	0.96	-1.90	0.85	-2.12	
surface-*OOH $\rightarrow$ HO*-surface-O*	0.99	-1.20	1.02	-1.23	0.91	-1.37	
$HO*$ -surface- $O* \rightarrow surface$ - $O* + H2O$	1.10	-1.48	1.12	-1.52	1.00	-1.70	
$surface-O^* \rightarrow surface-*OH$	1.03	-1.73	1.06	-1.77	0.94	-1.97	

Reaction steps of ORR pathways on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages

The Eactivation and  $\Delta G_{reaction}$  of reactions of pathways ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> is examined [22]. The  $\Delta G$  of reaction of ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> is calculated:

$$\Delta G = \Delta H - T\Delta S + \Delta G_{U} + \Delta G_{pH}$$
 (4)

The  $\Delta G_U = -$  neU, n is electron numbers and U is electrode potential.

The E<sub>activation</sub> of reactions of ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> is calculated:

$$E_{activation} = E_{TS} - E_{IS}$$
 (5)

The  $E_{TS}$  and  $E_{IS}$  are total energy of transition state and initial sate of reaction steps of ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub>. The HMO-LUMO gap energy (E<sub>HLG</sub>) and charge transfer (q) for Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages and complexes of ORR species with Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages are calculated.

$$E_{HLG} = E_{HOMO} - E_{LUMO}$$
 (6)

The E<sub>HOMO</sub> and E<sub>LUMO</sub> are energy of HOMO and LUMO of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages and complexes of ORR species with Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages.

The adsorption of OOH on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages has higher E<sub>adsorption</sub> than O<sub>2</sub> adsorption. The dissociation of O2 on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages has high E<sub>activation</sub> values. The dissociation of O2 molecules on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> to O + O have high E<sub>activation</sub>. The hydrogen atom can react by OOH on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages and the OOH adsorption on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages is more favorable process than O<sub>2</sub> dissociation.

In mechanism 1, the \*OOH on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> is hydrogenated and H<sub>2</sub>O is released and \*O on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> is produced. The \*O on Cr-Si $_{60}$ , Cr-C $_{60}$  and Cr-Ge $_{60}$  is hydrogenated and \*OH on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> is formed. The \*OH on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> is

hydrogenated and H<sub>2</sub>O is separated. The ΔG<sub>reaction</sub> of creation of the second H<sub>2</sub>O on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages are -0.69, -0.60 and -0.62 eV. In pathway 2, the \*OOH on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> is hydrogenated and HO\*-\*OH and -\*OH are formed and H<sub>2</sub>O is released. In final step, the \*OH on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages is hydrogenated and H<sub>2</sub>O is released. The E<sub>activation</sub> of creation of the second H<sub>2</sub>O on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages are 1.02, 1.13 and 1.15 eV. In mechanism 3, \*OOH on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> is dissociated to \*OH and \*O. The \*OH is hydrogenated and H2O is formed on Cr- $Si_{60}$ , Cr- $C_{60}$  and Cr- $Ge_{60}$ . The \*O on Cr- $Si_{60}$ , Cr- $C_{60}$ and Cr-Ge<sub>60</sub> is hydrogenated and\*OH is formed. The \*OH on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> is hydrogenated and Eactivation of creation of \*OH on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages are 0.96, 1.05 and 1.08 eV.

In mechanisms 1, 2 and 3 the formation of -\*OH, creation of -\*OH and H<sub>2</sub>O and production of -\*O and H<sub>2</sub>O are rate-determining steps for ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub>. The  $\Delta E_{activation}$  of -\*OH formation on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> 1 is lower than -\*OH, -\*O and H<sub>2</sub>O in pathways 2 and 3. The  $\Delta G_{reaction}$  of -\*OH formation on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> 1 is higher than -\*OH, -\*O and H<sub>2</sub>O in pathways 2 and 3. The pathway 1 is effective mechanisms for ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub>. The overpotential of ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> are lower than metal-based catalysts. The Cr-Si<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages have lower over-poential for ORR than Cr-C<sub>60</sub> nanocage. The Cr-Si<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages are catalyzed the ORR with high efficiency.

Table-2: The over-potential in V, E<sub>HLG</sub> in eV and q in e of complexes of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> with ORR species.

			l, 2p) model in gas				
	$\mathrm{Cr} ext{-}\mathrm{C}_{60}$		Cr-Ge <sub>60</sub>		Cr-Si <sub>60</sub>		
Complexes	$\mathbf{E}_{\mathbf{HLG}}\left(\mathbf{eV}\right)$	<b>q</b> (e)	$\mathbf{E}_{\mathbf{HLG}}\left(\mathbf{eV}\right)$	<b>q</b> (e)	$\mathbf{E}_{\mathbf{HLG}}\left(\mathbf{eV}\right)$	<b>q</b> (e)	
O-nano	-4.30	-0.23	-3.84	-0.26	-3.34	-0.30	
O-nano-O	-2.13	-0.47	-1.90	-0.53	-1.66	-0.60	
O-nano-OH	-2.42	-0.41	-2.16	-0.46	-1.88	-0.53	
HO-nano-OH	-2.49	-0.40	-2.22	-0.45	-1.93	-0.52	
O <sub>2</sub> -nano	-3.37	-0.03	-3.01	-0.03	-2.62	-0.04	
HOO-nano	-4.42	-0.11	-3.94	-0.13	-3.43	-0.15	
HO-nano	-4.65	-0.22	-4.15	-0.24	-3.61	-0.28	
H <sub>2</sub> O-nano	-3.29	-0.02	-2.94	-0.02	-2.55	-0.02	
Nanocages	Cr-Si <sub>6</sub>	0	$Cr-C_{60}$		Cr-Ge <sub>60</sub>		
Over-potential	0.25		0.29		0.33		
	B3LY	TP/6-311+G (2d	l, 2p) model in gas p	ohase			
	Cr-C <sub>6</sub>	0	Cr-Ge	60	Cr-Si <sub>60</sub>		
Complexes	E <sub>HLG</sub> (eV)	q (e)	$E_{HLG}(eV)$ q(e)		E <sub>HLG</sub> (eV)	q (e)	
O-nano	-4.11	-0.22	-3.67	-0.25	-3.19	-0.29	
O-nano-O	-2.04	-0.45	-1.82	-0.50	-1.58	-0.58	
O-nano-OH	-2.31	-0.39	-2.07	-0.44	-1.80	-0.51	
HO-nano-OH	-2.38	-0.38	-2.12	-0.43	-1.85	-0.50	
O <sub>2</sub> -nano	-3.23	-0.03	-2.88	-0.03	-2.50	-0.04	
HOO-nano	-4.22	-0.11	-3.77	-0.12	-3.28	-0.14	
HO-nano	-4.45	-0.21	-3.97	-0.23	-3.45	-0.26	
H <sub>2</sub> O-nano	-3.15	-0.01	-2.81	-0.02	-2.44	-0.02	
Nanocages	Cr-Si <sub>6</sub>	0	Cr-C <sub>6</sub>	50	Cr-Ge <sub>60</sub>		
Over-potential	0.23			0.27		0.31	
•		COSMO r	nodel in water				
	Cr-C <sub>6</sub>		Cr-Ge	60	Cr-Si <sub>60</sub>		
Complexes	E <sub>HLG</sub> (eV)	q (e)	E <sub>HLG</sub> (eV)	q (e)	E <sub>HLG</sub> (eV)	q (e)	
O-nano	-4.50	-0.24	-4.01	-0.27	-3.49	-0.31	
O-nano-O	-2.23	-0.49	-1.99	-0.55	-1.73	-0.63	
O-nano-OH	-2.53	-0.43	-2.26	-0.48	-1.96	-0.56	
HO-nano-OH	-2.60	-0.42	-2.32	-0.47	-2.02	-0.54	
O <sub>2</sub> -nano	-3.53	-0.03	-3.15	-0.03	-2.74	-0.04	
HOO-nano	-4.62	-0.12	-4.12	-0.13	-3.58	-0.15	
HO-nano	-4.86	-0.22	-4.34	-0.25	-3.77	-0.29	
H <sub>2</sub> O-nano	-3.44	-0.02	-3.07	-0.02	-2.67	-0.02	
Nanocages	Cr-Si <sub>6</sub>			Cr-C <sub>60</sub>		Cr-Ge <sub>60</sub>	
Over-potential	0.26			0.28		0.32	

#### Conclusion

In this work, the ORR mechanisms on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages are investigated. The Cr atoms of Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> are active positions to adsorb the ORR species. The adsorption OOH on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages has higher E<sub>adsorption</sub> than O<sub>2</sub> adsorption and dissociation of O<sub>2</sub> molecules on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages has high Eactivation values. The ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages is processed by three pathways. The  $\Delta E_{activation}$  of -\*OH formation on  $\text{Cr-Si}_{60}, \text{Cr-C}_{60}$  and  $\text{Cr-Ge}_{60}$  1 is lower than -\*OH, -\*O and H<sub>2</sub>O in pathways 2 and 3. The Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> have higher catalytic activity for ORR processes than metal-based catalysts and AlN, C, BN, AlP and BP nanocages and AlN, C, BN, Si, Ge, AlP and BP nanotubes. The  $\Delta G_{reaction}\, of$  -\*OH formation on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> 1 is higher than -\*OH, -\*O and H<sub>2</sub>O in pathways 2 and 3. The pathway 1 is acceptable mechanism for ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub>. The over-potential of ORR on Cr-Si<sub>60</sub>, Cr-C<sub>60</sub> and Cr-Ge<sub>60</sub> are lower than metal-based catalysts. The Cr-Si<sub>60</sub> and Cr-Ge<sub>60</sub> nanocages are proposed to catalyze the ORR with high efficiency.

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### References

- 1. Z. L. Zhao, O. Wang, L. Zhang, H. M. An, L. Zhou, M. L. Chang, Galvanic exchange-formed ultra-low Pt loading on synthesized unique porous Ag-Pd nanotubes for increased active sites toward oxygen reduction reaction, Electrochim. Acta, **263**, 209 (2018).
- 2. J. Yao, C. Li, K. Sun, Y. Cai, H. Li, W. Ouyang, H. Li, Ndc-scene: Boost monocular 3d semantic scene completion in normalized coordinates space. IEEE/CVF Int. Computer Vision 11, 9455 (2023).
- P. Li, J. Abbas, D. Balsalobre-Lorente, O. Wang, Q. Zhang, S. A. R. Shah, Impact of sectoral mix on environmental sustainability: How heterogeneity addressed? Gondwana Res. 128, 86 (2024).
- 4. J. X. Chen, L. Peng, J. Ma, H. P. Ying, Liberation of a pinned spiral wave by a rotating electric pulse. Euro. phys. Lett., 107, 38001 (2014).
- 5. X. Sun, S. Zhu, J. Guo, S. Peng, X. Qie, Z. Yu, P. Li, Exploring ways to improve China's ecological well-being amidst air pollution challenges using

- mixed methods. J. Environ. Management, 364, 121457 (2024).
- M. M. Khotbehsara, M. Zadshir, B. M. Miyandehi, E. Mohseni, S. Rahmannia, S. Fathi, Rheological, mechanical and durability properties of self-compacting mortar containing nano-TiO2 and fly ash. J. American Sci., 10, 222 (2014).
- 7. B. Mehdizadeh, K. Vessalas, B. Ben, A. Castel, S. Deilami, H. Asadi, Advances in Characterization of Carbonation Behavior in Slag-Based Concrete Using Nanotomography. Int. Variability, 1, 297 (2022).
- 8. B. Mehdizadeh Miyandehi, K. Vessalas, A. Mortazavi, Castel, M. Investigation Carbonation Behaviour in High-Volume GGBFS Concrete for Rigid Road Pavements. ASCP 1, 12
- 9. A. Naseri, B. Maleki, T. Asheghi Mehmandari, A. Tohidi, A. Fahimifar, Investigating the influence of Sample geometric variations on mechanical characterization in rock and concrete. J. Mining Environ., 16, 1089 (2025).
- 10. P. Jafari, E. Rasekh, T. Asheghi Mehmandari, M. Mohammadifar, A. Fahimifar, D. Armaghani, Upper-bound solutions for active face failure in shallow rectangular tunnels in anisotropic and non-homogeneous undrained clays. Geotech. Geolog. Eng., 43, 1 (2025).
- 11. T. A. Mehmandari, M. Shokouhian, M. Imani, A. Fahimifar, Experimental and numerical analysis of tunnel primary support using recycled, and hvbrid fiber reinforced shotcrete. In Structures, 63, 106282 (2024).
- 12. Mehmandari, T. A., Shokouhian, M., Imani, M., Tee, K. F., Fahimifar, A. Split Tensile Behavior of Recycled Steel Fiber-Reinforced Concrete. ACI Mat. J., 122, 5147 (2025).
- 13. T. A. Mehmandari, M. Shokouhian, M. Z. Josheghan, S. A. Mirjafari, A. Fahimifar, D. J. Armaghani, K. F. Tee, Flexural properties of fiber-reinforced concrete using hybrid recycled steel fibers and manufactured steel fibers. J. Building Eng., 98, 111069 (2024).
- 14. T. A. Mehmandari, D. Mohammadi, M. Ahmadi, M. Mohammadifar, Fracture mechanism and ductility performances of fiber reinforced shotcrete under flexural loading insights from digital image correlation (DIC). Insight Civil Eng., 7, 611 (2024).
- 15. M. Ershadi, M. Javanbakht, D. Brandell, S. A. Mozaffari, A. M. Aghdam, Facile synthesis of amino-functionalized mesoporous Fe3O4/rGO 3D nanocomposite by diamine compounds as Liion battery anodes. Appl. Sur. Sci., 601, 154120 (2022).

- 16. A. Molaei Aghdam, S. Habibzadeh, M. Javanbakht, M. Ershadi, M. R. Ganjali, High interspace-layer manganese selenide nanorods as a high-performance cathode for aqueous zinc-ion batteries. ACS Appl. Energy Mat., 6, 3225 (2023).
- 17. A. Molaei Aghdam, N. Mikaeili Chahartagh, E. Delfani, High-Efficient Capacitive Deionization Using Amine-Functionalized ZIF-67@2D MXene: Toward Ultrahigh Desalination Performance. Adv. Mat. Techn., 8, 2300628 (2023).
- 18. A. M. Aghdam, N. M. Chahartagh, S. Namvar, M. Ershadi, F. B. Ajdari, E. Delfani, Improving the performance of a SnS 2 cathode with interspace layer engineering using a Na+ insertion/extraction method for aqueous zinc ion batteries. J. Mat. Chem. A, 12, 1047 (2024).
- 19. F. B. Ajdari, F. Abbasi, A. M. Aghdam, F. G. C. Khaneh, A. G. Arjenaki, V. Farzaneh, S. Ramakrishna, Innovative self-repairing binders tackling degradation and de-lithiation challenges: Structure, mechanism, high energy and durability. Mat. Sci. Eng. Reports, 160, 100830 (2024).
- 20. Y. Ru, M. Gruninger, Y. Dou, Robust selfsupervised symmetric nonnegative matrix factorization to the graph clustering. Sci. Reports, 15, 7350 (2025).
- 21. Y. Huo, S. Gang, C. Guan, Fcihmrt: Feature cross-layer interaction hybrid method based on Res2Net and transformer for remote sensing scene classification, Electronics, 12, 4362 (2023).
- 22. J. Singh, R. Abraham, Magnetic Catalyst CdFe2O4: Direct Conversion of Thiols into Antibacterial Sulfonamides. Biol. Mol. Chem., 2, 13 (2024).